

# THE INFLUENCE OF DEPOSITION TEMPERATURE ON VANADIUM DIOXIDE THIN FILMS MICROSTRUCTURE AND PHYSICAL PROPERTIES

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## ABSTRACT

Vanadium dioxide thin films were successfully prepared on soda lime glass substrates using the optimised conditions for r.f.-inverted cylindrical magnetron sputtering. The optimised deposition parameters were fixed and then a systematic study of the effect of deposition temperature, ranging from 450 °C to 550 °C, on the microstructure of thermochromic thin films was carried out. The deposited films were found to be well crystallised, showing strong texture corresponding to the (011) plane, indicating the presence of vanadium dioxide.

## INTRODUCTION

Vanadium dioxide (VO<sub>2</sub>) is an interesting material because of its transition from semiconducting to metallic behaviour at a transition temperature of about 70 °C.<sup>1</sup> This behaviour is accompanied by a microstructural change from a monoclinic (below transition temperature) to a tetragonal structure (above transition temperature).<sup>2</sup> VO<sub>2</sub> has a narrow band gap (0.65 eV) below the transition temperature.<sup>3,4</sup> Numerous researchers<sup>5,6,7</sup> have reported that VO<sub>2</sub> phase change is accompanied by an abrupt change in the electrical resistance and optical transmittance. It has low transmittance around the infrared region during the metallic phase and has higher transmittance in the infrared region during the semiconducting phase. As a result of this behaviour, VO<sub>2</sub> has been of interest for various applications, such as the 'smart window', laser protection and electrical auto-switching. The transition temperature (~70 °C) can be decreased or increased using dopants (e.g. tungsten molybdenum) depending on the purpose of the application. The VO<sub>2</sub> film thickness decreases the transition temperature and hence the optical transmittance decreases.<sup>8</sup> In this paper, we report on the effect of deposition temperature on the microstructure of VO<sub>2</sub> thin films. The VO<sub>2</sub> thin film microstructure was characterised using ultraviolet-visible (UV-VIS) spectrophotometry and X-ray diffraction.

## EXPERIMENTAL SET-UP

The films were deposited on soda lime glass substrates using r.f.-inverted cylindrical magnetron sputtering (ICMS) from a vanadium target in an argon and oxygen atmosphere (Figure 1). This technique is defined as a supplementary exceptional way to engineer nano-systems at high deposition rates onto complex-shaped substrates. This system consists of a cylindrical ring vanadium metal target with a circular magnet enclosed behind the target. The ring magnets are there to enhance the electron path. Because the electron path is not on a straight line as a result of the magnetic field, the collision probability of an electron with ions is high and therefore the plasma is enhanced. The vacuum chamber was fitted with two gas inlets to introduce a mixture of argon and oxygen gases for reactive sputtering. The gas flow was controlled with a leak valve and the pressure inside the chamber was monitored by Penning and Pirani gauges. A Dressler Cesar RF power generator was used as a power supply. Prior to the deposition, the system was evacuated by a diffusion pump backed by a rotary pump down to 10<sup>-6</sup> mbar. A mixture of argon and oxygen (argon was used as the working gas for enhancing plasma and oxygen as a reactive gas), was let into the sputter chamber. For all deposition processes, a mixture of argon (90%) and oxygen (10%) was introduced to the chamber at a pressure of 8x10<sup>-3</sup> mbar followed by a high-purity argon (99.99%) flow at a pressure of 10<sup>-2</sup> mbar. This gas composition and pressure were found to be optimal for getting thermochromic VO<sub>2</sub>. The deposition time and power were fixed at 2 h and 70 W, respectively, in all depositions. The deposition temperature ranged from 450 °C to 550 °C. The lowest deposition temperature to obtain crystalline VO<sub>2</sub> was found to be 450 °C.

## SAMPLE CHARACTERISATION

The following sections describe the characterisation technique used and the conditions during experimentation.

### X-ray diffraction technique

X-ray diffraction (XRD) is described as a powerful non-destructive technique for the characterisation of crystalline materials.<sup>9,10</sup> It provides information on the structure, phase, preferred crystal orientation (texture) and other structural parameters, such as average grain size, strain and crystal defects.

When a monochromatic X-ray beam with wavelength  $\lambda$ , on the order of lattice spacing  $d$ , is projected onto a crystalline material at an angle  $\theta$ , X-ray diffraction peaks are produced by constructive interference of the monochromatic beam scattered from each set of lattice planes at specific angles. Constructive interference gives the diffraction peaks according to Bragg's law,

$$2d \sin \theta = n\lambda. \quad [\text{Eqn 1}]$$

By varying the angle  $\theta$ , the Bragg's law condition is satisfied by different  $d$ -spacings in polycrystalline materials. Plotting the angular position and intensities of the resultant diffracted peaks of radiation produces a pattern, which is characteristic of the material. The full width at half maximum of the peak,  $\Delta(2\theta)$  (in radians), is a measure of the grain size  $b$  in a polycrystalline film or the mosaic blocks in an epitaxial layer, as described by Scherrer's formula:

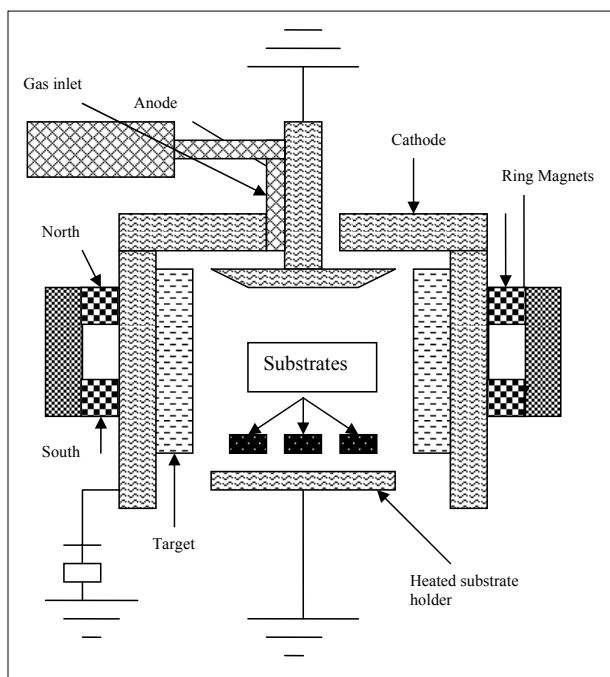


FIGURE 1

A schematic illustration of an r.f.-inverted cylindrical magnetron sputtering system

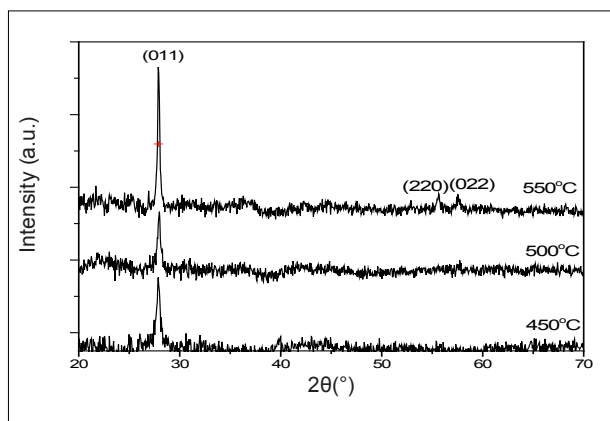


FIGURE 2

X-ray diffraction patterns for VO<sub>2</sub> deposited by r.f.-inverted cylindrical magnetron sputtering at different deposition temperatures

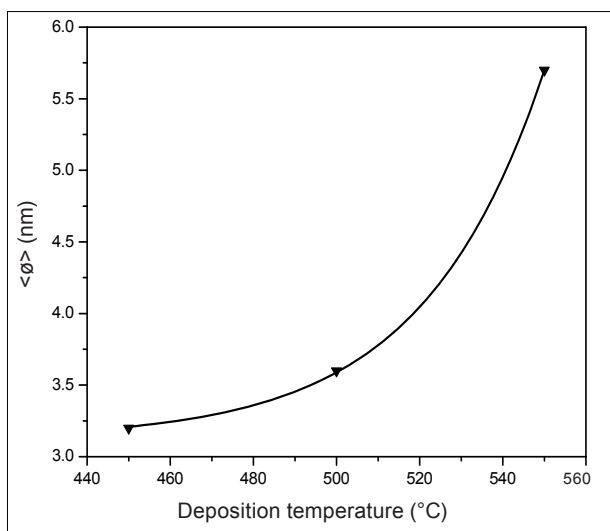


FIGURE 3

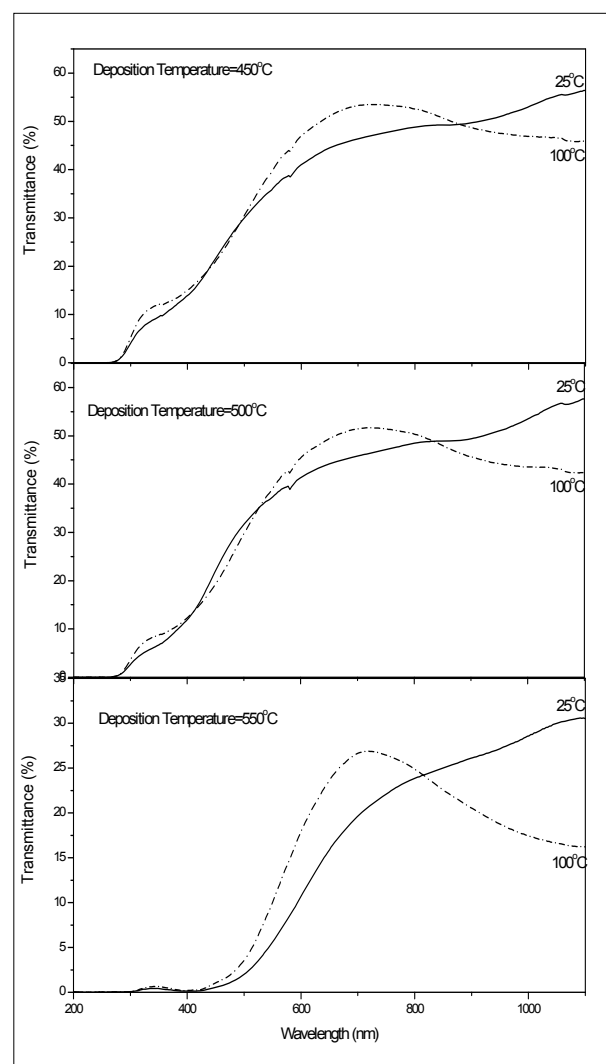
A plot showing the effect of deposition temperature (°C) on grain size (nm)

TABLE 1  
The reflection planes of VO<sub>2</sub>, with corresponding values of 2θ, obtained by X-ray diffraction

2θ (°)	Phase	Plane
27.72	VO <sub>2</sub>	(011)
57.63	VO <sub>2</sub>	(220)
65.61	VO <sub>2</sub>	(022)

TABLE 2  
Transition temperature as a function of the deposition temperature

Deposition temperature (°C)	Transition temperature (°C)
450	66
500	70
550	65



Note: The solid and dashed lines symbolise the semiconducting and metallic states of the films, respectively.

FIGURE 4

Optical transmission (%) as a function of wavelength (nm) for VO<sub>2</sub> thin films deposited by r.f.-inverted cylindrical magnetron sputtering for a period of 2 h at different deposition temperatures

$$b = \frac{8.89\lambda}{\Delta(2\theta) \cdot \cos(\theta)} \quad [\text{Eqn 2}]$$

The XRD equipment that was used for this work is the Bragg-Brentano diffractometer, with Cu K<sub>α</sub> (λ = 0.15418 nm) radiation and a scintillation detector. The X-ray tube was typically operated at a voltage of 40 kV and a current of 20 mA. One scan mode, θ – 2θ scan, was used for microstructural characterisation. The detector rotated twice as fast and in the same direction

around the diffractometer axis as the sample. This technique is also called the locked-coupled scan. In the  $\theta - 2\theta$  scan, the reflections from the planes parallel to the substrate surface were detected. This allowed us to determine the orientation along the growth direction of the (epitaxial or polycrystalline) films with respect to the substrate.

### UV-VIS spectrophotometry

The UV-VIS spectrophotometer that was used in this work consists of a light source, a sample holder fitted with a heating stage, a detector and a data acquisition computer.<sup>11,12</sup> The sample holder is located between the light source and the detector. The light source, detector and sample holder are designed to fit within a dark box that has a cover to prevent unwanted light interfering with the laser beam during experimentation. The wavelength range of the spectrophotometer is 200 nm – 1100 nm. The spectrophotometer was allowed to create a baseline which removes background from the data to be collected. The beam of light is directed to the sample and the detector detects the amount of light transmitted through the sample.

The data was generated and captured by the computer and temperature measurements were recorded. The data collected was used to plot the graphs of transmittance as a function of wavelength. The UV-VIS spectra were used to investigate the optical switching property of the deposited VO<sub>2</sub> films.

## RESULTS AND DISCUSSION

### Film microstructure

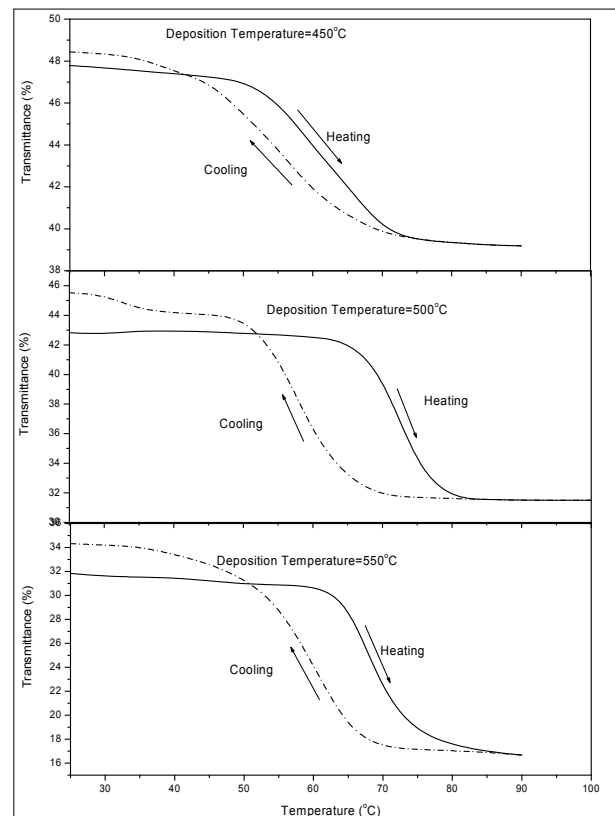
The XRD spectra (scanned from 20° to 70°) of the VO<sub>2</sub> thin films deposited at different deposition temperatures are shown in Figure 2. The scan was taken at room temperature, in the semiconducting phase of VO<sub>2</sub> (monoclinic structure). The films exhibit a polycrystalline structure whose peaks match the single phase of monoclinic VO<sub>2</sub> (JCPDS 044-0225) without showing the indication of other VO<sub>2</sub> phases. The reflection planes with their corresponding values of  $2\theta$  are given in Table 1. All the films show a first peak at the (011) plane, indicating the presence of VO<sub>2</sub>.<sup>6,13,14,15</sup> The XRD spectra of the film deposited at a deposition temperature of 450 °C shows Bragg reflection at  $2\theta = 27.72^\circ$ , corresponding to (011) reflection of the VO<sub>2</sub> monoclinic phase. The grains of the VO<sub>2</sub> thin films deposited at 450 °C are highly oriented with crystallographic (0hh) planes parallel to the substrate plane. The (011) reflection emerges from a broad structure. The broad structure may suggest the existence of a disordered phase of VO<sub>2</sub>. The XRD spectra of the film deposited at a deposition temperature of 500 °C exhibited the same behaviour as that of the film deposited at 450 °C. The film deposited at a deposition temperature of 550 °C exhibited similar XRD patterns. In addition to the (011) peak, the patterns show the emergence of new peaks at  $2\theta = 57.63^\circ$  and  $65.61^\circ$  corresponding to the (220) and (022) reflections, respectively. The appearance of new reflections is a clear indication of the deterioration of the orientation of the VO<sub>2</sub> grains when the temperature increases. The peak at the (011) reflection was used to estimate the average grain size. The average grain size was estimated using Debye's formula:

$$\langle \sigma \rangle = \frac{0.9\lambda}{FWHM \cos\theta} \quad [\text{Eqn 3}]$$

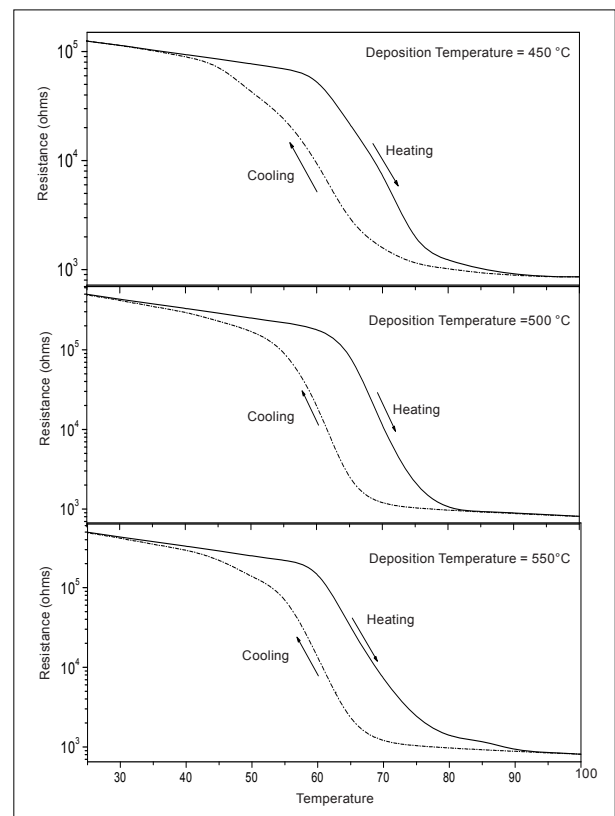
The estimated values were then used to plot the graph shown in Figure 3. The graph was used to observe the influence of deposition temperature on the grain size of the VO<sub>2</sub> thin film.

### Thermochromism of the film

Figure 4 shows the film optical transmittance (%) as a function of wavelength (nm) of the VO<sub>2</sub> thin films deposited by ICMS. The spectral transmittance observed at a low temperature (25 °C) semiconducting phase and a high temperature (100 °C) metal phase showed a sharp contrast in the infrared region. The



**FIGURE 5**  
Optical transmission (%) as a function of temperature (°C) at a wavelength of 1100 nm for VO<sub>2</sub> thin films deposited by r.f.-inverted cylindrical magnetron sputtering at different deposition temperatures



**FIGURE 6**  
Resistance (ohms) as a function of temperature (°C) at a wavelength of 1100 nm for VO<sub>2</sub> thin films deposited by r.f.-inverted cylindrical magnetron sputtering at different deposition temperatures

metallic phase exceeds the semiconducting phase in the infrared region in all films, which indicates that the film is less than 100 nm as demonstrated by Xu et al.<sup>8</sup> The highly oriented VO<sub>2</sub> film obtained at 450 °C exhibits higher transmission switching efficiency than the films obtained at 500 °C and 550 °C. Figure 5 show the hysteresis loops of transmittance as a function of temperature (at a fixed wavelength of 1100 nm) for the VO<sub>2</sub> thin films. The hysteresis width of the films deposited at a deposition temperature of 450 °C is approximately 5 °C. The films obtained at 500 °C and 550 °C have the same hysteresis width of approximately 10 °C. The transition temperatures are shown in Table 2. All the films have the same hysteresis sharpness. The transition temperature of the films slightly decreases when deposition temperature increases. It is suggested that this slight decrease is caused by the random arrangement of the grains in the film deposited at 550 °C.

### Electrical measurements of the film

Figure 6 shows the resistance hysteresis loops of VO<sub>2</sub> thin films deposited at temperatures of 450 °C, 500 °C and 550 °C. The film deposited at 450 °C exhibits a sharp transition in electrical resistance near 68 °C, which is a definitive confirmation of formation of thermochromic VO<sub>2</sub>. The drop in the electrical resistance is about two orders of magnitude, which is similar to the one observed in high-quality polycrystalline VO<sub>2</sub> thin films but less than that reported by Kucharczyk and Niklewski<sup>16</sup> for a single bulk crystal of VO<sub>2</sub>. The temperature hysteresis width is approximately equal to 10 °C. This large hysteresis of the VO<sub>2</sub> film compared to that observed in a single bulk VO<sub>2</sub> crystal (2 °C), may have its origin in the extrinsic stress created during deposition by the difference in thermal expansion coefficients between the VO<sub>2</sub> film and the substrate. The films deposited at 500 °C and 550 °C exhibit the same behaviour in terms of electrical switching. The transition is sluggish and shifts towards lower temperature near 66 °C. The drop in the electrical resistance is about three orders of magnitude. Both films exhibit a temperature hysteresis width of about 10 °C. The decrease in the sharpness of the transition profile and the magnitude of the drop in the electrical resistance is caused by the random arrangement of the grains in the film, leading to different grain boundaries. The increase in deposition temperature affected the microstructure of the film, which led to the decrease in electrical and optical properties.

### CONCLUSION

VO<sub>2</sub> thin films were successfully synthesised on soda lime glass substrates. An increase in deposition temperature gives rise to an increase in grain size and causes deterioration of the orientation (texture) of the grain. An increase in deposition temperature enhances the peak intensity. An increase in deposition temperature also results in decreases in the optical and electrical properties of the film. It is clear that the electrical and optical switching of VO<sub>2</sub> thin films depends on the synthesis conditions of the technique.

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